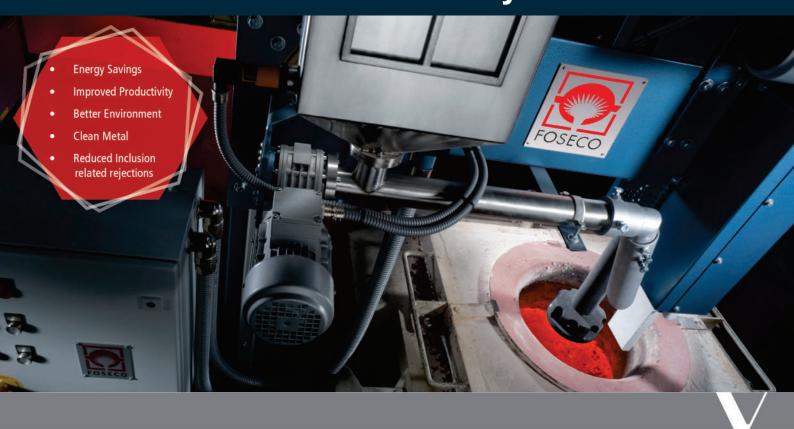


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JOURNAL FOR ALUMINIUM CASTING TECHNOLOGY

Volume 56 - February 2023



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We always look forward to your Feedback and comments on the Journal. Please do write to us.

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VESUVIUS



FROM THE FOSECO ARCHIVES

THEORY AND PRACTICE OF GRAIN REFINING FOR ALUMINIUM ALLOYS - UTILIZING COVERAL MTS 1582

Authors: Brian Began, Foseco USA & Pascaline Careil, Foseco Europe



The need for smaller grains is vital to achieving the required properties when pouring most cast aluminum alloys. Whether the desired results are high mechanical properties, leaker free castings, a cosmetic appearance or improved structural soundness, smaller grains are impactfully beneficial. Accordingly, there is a desire to improve both grain refining and the ability to quickly and effectively assess grain refinement effectiveness. This paper discusses both the need for smaller grains and the principle fundamentals of grain refining. Moreover, the paper reviews commercially-available grain refiner forms and currently available methods for assessing grain refinement. Finally, the paper introduces a new and improved flux form grain refiner (COVERAL MTS 1582) and documents two recently successful case studies where the COVERAL MTS 1582 was utilized to improve castings in both a low pressure wheel foundry and a high production sand moulding foundry, respectively.



INTRODUCTION

Grain refining is an essential part of the aluminium casting process which aims at reducing the size of primary aluminium grains during the solidification phase. This process has many benefits for most hypoeutectic aluminium alloys as it improves feeding, elongation and mechanical properties, increases resistance to fatigue, improves casting machinability, reduces hot tears, helps disperse micro-shrinkage, decreases the size of porosities and reduces thermal treatment cycles. Historically, grain refinement has been achieved using master alloys, with the most commonly used grain refiner mechanism involving the release of Titanium diboride into the melt. Grain refining is especially important in aluminium foundries using investment, sand, gravity die,

or low pressure die casting fatigue properties. processes due to the potential for delayed cooling and complex casting designs with varying section thickness.

In general, those castings with slower cooling rates and larger variation in casting thickness, require grain refinement more than other casting designs.

There are several casting segments where grain refining is critical

- Wheel foundries where grain refinement and cleaning are crucial for achieving the required feeding and cosmetic surface finished of the
- + Safety critical automotive castings such as suspension parts, turbochargers, and brake components which require good

- General automotive castings like cylinder heads, engine blocks, manifolds in gravity diecast where an intermediate level of grain refinement might suffice for the mechanical property requirements, but the improved feeding from grain refinement helps prevent leakers.
- Aerospace and military castings requiring high mechanical properties for difficult applications, grain refining is highly beneficial.
- + Sand and investment castings where the long solidification times cause large grain growth and difficult feed paths without optimized grain refining.

GRAIN REFINEMENT MECHANISM IN ALUMINIUM ALLOYS

TARGET OF ALL MELT TREATMENT PROCEDURES IS AN IMPROVEMENT OF MECHANICAL PROPERTIES

Grain refinement affects the α -mixed crystal in the alloy. At decreasing temperature those α -mixed crystals grow. Grain size depends on cooling rate during solidification. The addition or formation of nuclei increases solidification speed and decreases the grain size.

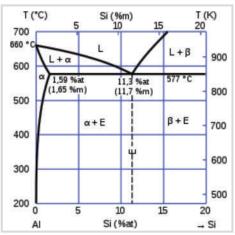


Figure 1: Al-Si phase diagram

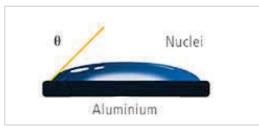
Ceramic	Angle
TiB	60°
ZrBį	106°
HfB	134°
ТаВ	125°
TiC	118°
ZrC	150°
SiC	135°
HfC	148°
NbC	136°
TaC	145°
TiN	135°
ZrN	167°
NbN	156°
AIN	138°

Table 1: Contact angle of different ceramic materials [1]

MASTER ALLOY AND CHEMICAL PRODUCTS COMPARISON

Considerations when using master alloy grain refiners

- + AlTi5B1 AlTi3B1 AlTi5B0,2 AlTi10B1
- + TiB₂ nuclei are pre-formed in an aluminium matrix
- + Easy to apply
- + Risk of oxides or impurities in the rod or waffle
- + Moisture and oxides on rod surface contaminate melt



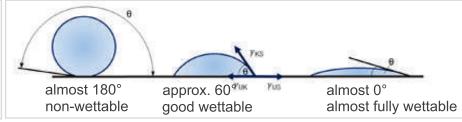


Figure 2: Nuclei needs a good wettability by melt Figure 3: Heterogeneous nucleation as a function of wetting angle

$$\sigma_{\alpha F} = \sigma_{\beta F} + \sigma_{\alpha \beta} \cos (\theta)$$

 $\sigma_{\alpha F}$ = melt surface energy

 $\sigma_{\rm BF}^{\rm m} = {\rm surface\ energy\ of\ nuclei}$

 $\sigma_{\alpha\beta}^{\prime\prime}$ = interface energy between nuclei and melt

Figure 4: Young's equation

Benefits of chemical products are:

- + Contain metallic titanium and boron salt or titanium and boron salt
- + TiB₂ nuclei are in-situ formed in the melt fresh surface higher surface energy and lower θ angle
- + No risk of impurities
- + Additional cleaning effect

REASONS FOR BETTER MECHANICAL STRENGTH WITH CHEMICAL PRODUCTS

We identified several reasons for archieving better mechanical strength with chemical products which are:

- + Chemical products and master alloys with preformed nuclei impact the contact angle θ differently
- + θ for TiB₂ = 60° is a theoretical value for an ideal nucleus
- + θfor TiB₂ from master alloys is significantly higher due to reduced surface energy
- + θfor TiB₂ from chemical products is close to 60° or even below due to fluxing effect from chemicals (fluorides)

COVERAL MTS 1582

FOSECO has developed a novel granulated flux COVERAL MTS 1582 that is capable of both grain refining and cleaning aluminium alloy melts. COVERAL MTS 1582 is highly concentrated in titanium and boron which form both titanium diboride and aluminum boride in situ leaving fresh nuclei within the aluminium melt. These finely dispersed species are highly efficient nuclei that promote a fine equiaxed grain growth during solidification.

In addition to strong grain refining, COVERAL MTS 1582 fluxis also a very good cleaning product that will react to remove oxides and inclusions from the melt. No additional cleaning/drossing flux is required, resulting in lower overall process costs.

COVERAL MTS 1582 is a sodium- and calcium-free granulated flux suitable for all types of aluminium alloys except hypereutectic alloys but including those alloys containing large amounts of magnesium.

APPLICATION OF COVERAL MTS 1582

COVERAL MTS 1582 is specially designed for use with FOSECOs MTS 1500 rotary degassing and melt treatment equipment, whereby controlled flux additions are made directly into a melt vortex and mixed vigorously. PLC controlled additions of the treatment flux are added into the vortex and mixed to complete reaction prior to the vortex breaker baffle board re-engaging the melt, effectively stopping the vortex. After the vortex has been stopped, the MTS completes a standard rotary degassing process and the treated metal in the ladle or crucible is used for transferring and/or casting.

For further information of the MTS 1500 process, readers are advised to review Foundry Practice Issue 247 (2007) or the Foundry Practice Special Edition for AFS CastExpo (2008).

Both issues feature excellent articles on the MTS 1500 technology [References 2 and 3].

MTS 1582 should be used with the melt at a temperaturehigher than 720 C. The reaction byproduct from this treatment produces an extremely dry ash-like dross that is easily separated from the liquid metal with a coated skimmer or similar tool.

EVALUATING GRAIN REFINEMENT EFFECTIVENESS

Since grain refinement is critical to achieve the desired properties of aluminium castings, it is important that there are methods for assessing grain refinement effectiveness. The most common methods for evaluating grain refinement effectiveness are as follows:

- + Elemental spectroscopy
- + Thermal analysis
- + Microstructural evaluation

ELEMENTAL SPECTROSCOPY

Elemental spectroscopy is perhaps the most commonly

employed method for assessing grain refinement, but it is also the least effective of the three methods listed.

Spectroscopy only determines the total concentration of an element - however Titanium is usually present in other forms and phases in addition to TiB_2 and these other phases do not impact grain structure. Foundries will measure Ti into the alloy range (typically 0.10-0.25% by weight) and assume that because they are in range, they are achieving sufficient grain refinement. Consequently, given this issue, some foundries will also measure boron (typical range 5-25ppm) as an additional control. Tight controls of Ti and B do typically result in effective grain refining; however, more advanced methods like thermal analysis and microstructural analysis ensure higher probabilities of optimized grain refinement.

THERMAL ANALYSIS

Thermal Analysis is perhaps the fastest growing method for assessing grain refinement as it is quick and more accurate than elemental spectroscopy. The THERMATEST* 5000 NG III (pictured in Figure 5) is a widely used thermal analysis unit used to quickly and accurately assess grain refinement effectiveness in aluminium alloys. Thermal analysis involves collecting data of temperature versus time of a solidifying melt sample and comparing the curve to a set of known reference curves algorithmically. The THERMATEST 5000 NG III unit's algorithm analyzes the sample curve liquidus and computes a score on a scale from 1-9 for evaluating grain fineness (GF). A score of 1 references a curve that compares with curves exhibiting no grain refining.

In contrast, a GF score of 9 is achieved when the sample curve compares with those curves known to have produced "perfect" grain refining of melts with

the same alloy composition. A pictorial representation of the THERMATEST 5000 NG III grain refinement levels is provided in Figure 7. Of note, THERMATEST 5000 NG III unit also provides the side benefit of helping to assess eutectic modification effectiveness in Al-Si alloys [References 4 and 5].

EVALUATION OF GRAIN SIZE WITH THERMAL ANALYSIS

For a given cooling speed, the size of the grain depends on the amplitude and duration of the undercooling, which appears at the formation of primary aluminium crystals.

- + When the undercooling is high and duration medium
 - (Fig 6a), grain size is coarse.
- + When there is no undercooling (Fig 6b), grain size is very fine.
- + When undercooling is low but duration is high, the grain size is very coarse.



Figure 5: Photograph of a THERMATEST 5000 NG III unit

THERMATEST 5000 NG III measures the following Liquidus

parameters:

- + Temperature θ2 (°C)
- + Undercooling Δθ(°C)
- + The duration of undercooling t1 (in seconds)

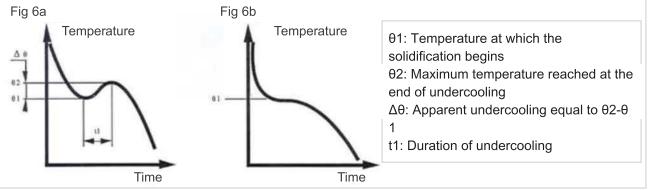


Figure 6a and b: Profiles of the cooling curve at the solidification of primary aluminium crystals in case of hypoeutectic alloy

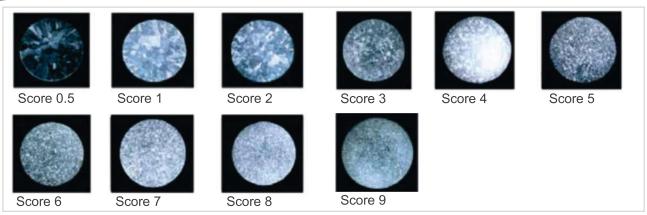


Figure 7: Test of grain refinement - Standard plate with Grain Fineness (GF)

Grain refinement is considered fully optimized when the undercooling is nil and grain size index is equal to 9. However, for certain alloys and thin shaped castings in permanent moulds, a lower grain size index (5 - 9) would be expected and is acceptable due to the higher cooling rate with permanent die casting.

We recommend setting a minimum grain size index for each casting, correlated with desired elongation of mechanical properties. For Al-Cu5%MgTi alloys, the absence of undercooling may not be sufficient to avoid hot tears. A stronger grain refinement is recommended to improve the alloy's performance.

LIQUIDUS CURVES: COMPARISON OF TIB RODS WITH COVERAL MTS 1582

The lower the undercooling at Liquidus, the stronger the grain refinement. COVERAL MTS 1582, at much lower addition rate (0.11% vs 0.2% for AlTi5B1 rods), performs better compared to AlTi5B1 rods.

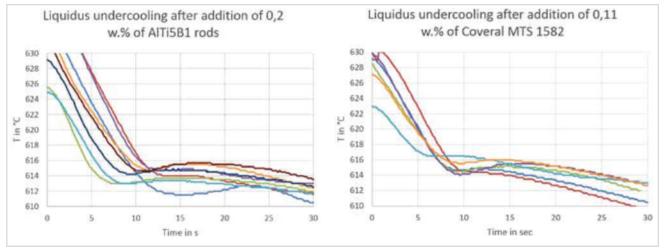


Figure 8: Thermal analysis curves

OPTICAL MICROSCOPY (BARKER TEST)

Optical microscopy is the final methodology employed by foundries to assess grain refinement. Optical microscopy is considered the most representative method for assessing grain refinement but is time-consuming and resource intensive. Optical microscopy involves grinding and polishing test specimens to microscopic levels to be evaluated for grain size under a

microscope. One popular method for optical microscopy is the Barker test. The LectroPol-5 from

Struers is used for electrolytic etching with Barker reagent consisting of a 5% tetrafluoroboric acid in distilled water. The sample to be tested acts as an anode in a galvanic cell, which removes material from the sample surface and an anodic layer can be formed. With the Barker method, under a polarized light, a colored representation of the

grain structure of aluminum materials is achieved. It is possible to carry out microscopic testing with up to 1000x magnification.

Alloy: AlSi7Mg0,3 COVERAL MTS 1582 Addition rate: 0,1%



Figure 9a: Before treatment. Grain size dm $[\mu m] = 984$

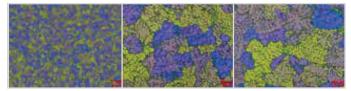
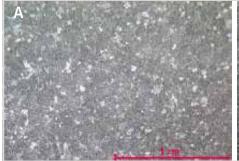


Figure 9b: After treatment. Grain size dm $[\mu m] = 206$





- A Alloy: AlSi7Mg0,3 Rods AlTi5B1 0.08 % addition rate Grain size $\emptyset = 422 \mu m$
- **B** COVERAL MTS 1582 0.05 % addition rate Grain size $\emptyset = 237 \mu m$

Figure 10: Comparison of TiB rods with Coveral MTS 1582: grain size

CASE STUDIES

WITH COVERAL MTS 1582

1: European foundry

A European wheel foundry was interested in improving its melt treatment practices by utilizing COVERAL MTS 1582 with a FDU featuring MTS 1500 technology. This wheel foundry pours a standard AlSi7Mg alloy and historically performed grain refining by making manual additions of TiBor rod into a transfer ladle during degassing. It was the foundry's target to automate the grain refining process all while capturing the typical benefits (drier dross, lower spend, smaller grain) achieved when grain refining with COVERAL MTS 1582. The treatment parameters of the new process featuring COVERAL MTS 1582 can be found in Table 2.

After the new process grain refining with COVERAL MTS 1582 was implemented, pictures were taken of the ladle dross (Figure 11), thermal analysis curves (Figure 13) and microstructures (Figure 12).

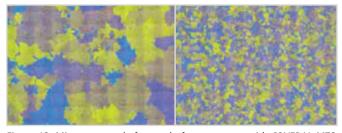


Figure 12: Microstructure before and after treatment with COVERAL MTS 1582.

Treatment parameters				
Ladle	INSURAL ATL 600 with 500 kg of AlSi7Mg			
Temperature	730 - 760 °C			
Addition rate	250 g COVERAL MTS 1582 (0.05 % of the melt weight)			
Treatment time	6 minutes			
Inert gas flow	20 l/min N ₂			
Rotor speed	450 rpm for MTS FDR 190.70			

Table 2: European Wheel Foundry (EWF) treatment parameters.

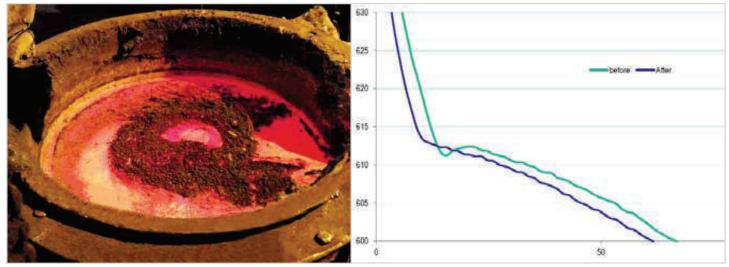


Figure 11: Photograph of extremely dry dross in transfer ladle after treatment with COVERAL MTS 1582.

Figure 13: Thermal analysis curves

2: American foundry

Littlestown Foundry is a sand and low-pressure (LP) mould aluminium foundry in Littlestown, Pennsylvania in the USA. The main alloy poured by Littlestown Foundry is a standard 356 alloy (AlSi7Mg). In the sand foundry, Littlestown casts some difficult castings that are subjected to pressure testing with air to make sure they are leak free for application. After reducing scrap through improved grain refining in the LP foundry from 13.6% to 2.7% by converting from metallic TiBor (10%Ti, 1%B) to COVERAL MTS 1582, a similar project was undertaken in the sand foundry. The aim was that by improving the grain refining using COVERAL MTS 1582 - introduced via an MTS 1500 unit - in place of metallic TiBor rod, the sand foundry would see similar benefits in the form of reduced leakers and lower spend.

The first part of the project involved using a THERMATEST 5000 NG III unit to assess the incumbent procedure and then developing an optimized procedure using the MTS 1500 and COVERAL MTS 1582. The results of the THERMATEST 5000 NG III evaluation are presented in Table 3.

Sample #	Average Grain Fineness (GF)		
Sample before treatments	5.8		
Standard TiBor Additions	6.8		
COVERAL MTS 1582	9.0		

Table 3: Results of THERMATEST 5000 NG III evaluation with the COVERAL MTS 1582 grain refining flux.

The THERMATEST 5000 NG III evaluation confirmed that the metallic TiBor rod was successful in raising the Grain Fineness value from insufficient (5.8/9.0) to an improved and more acceptable level of grain refining (6.8/9.0). However, the THERMATEST 5000 NG III unit also confirmed that a huge improvement to a fully optimized level of perfect grain refinement (9.0/9.0) was possible with the COVERAL MTS 1582. Hence, mechanical test bars were poured and evaluated to assess any potential impact of the new process featuring the COVERAL MTS 1582.

The results of the mechanical testing evaluation are shown in Table 4. The results exhibited positive improvement in all three metrics evaluated, i.e., ultimate tensile strength (UTS), yield strength (YS) and % Elongation. Accordingly, the decision was made to convert to the new process to make a full assessment of the new process featuring COVERAL MTS 1582 and a FDU featuring MTS 1500 technology.

Test Incumbent TiBor Proces		New Process Featuring MTS 1500 & COVERAL MTS 1582
UTS in psi (MPa)	40,000 (276)	41,290 (285)
YS in psi (MPa)	34,500 (238)	35,100 (242)
Elongation (%)	4%	5%

Table 4: Results of mechanical testing of preceding treatment samples and samples collected after the implementation of a MTS 1500 with COVERAL MTS 1582.

Finally, after four months in production, the new process change was evaluated economically. The following economic benefits were achieved after implementation:

- + Reduction in annual projected spend on grain refiners and cleaning flux by \$276 per day, \$1,380 per week, \$5,750 per month or more than \$69,000 per year.
- + A ten-fold reduction in projected impregnation costs from a starting point that exceeds \$1,500 per month to less than \$150 per month.
- + The calculated payback for the MTS 1500 unit when

factoring in the lower grain refining spend, the lower flux cleaning flux spend and offsetting it with the slightly

higher spend on filters is just over 6 months.

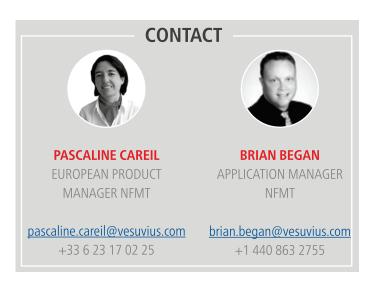
A full peer-reviewed paper (paper #19-015) on the Littlestown Case Study was published with the AFS 123rd Metalcasting Congress Proceedings in April 2019 and is available for a more extensive review.

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SUMMARY & CONCLUSIONS

COVERAL MTS 1582 is a universal grain refining and cleaning flux for treating aluminium alloys. It forms in situ Aluminium boride and Titanium boride which are the most suitable nuclei, within aluminium melts. Creating TiB2nuclei in situ is more effective than releasing pre-made TiB2nuclei into a melt. Elemental spectroscopy, thermal analysis with a THERMATEST 5000 NG III and optical microscopy are three methods for assessing grain refinement effectiveness within a melt; the latter two methods being the most efficient. Experience in both a low pressure wheel foundry and high production greensand foundry has confirmed the benefits of superior casting mechanical properties and lower overall process costs when grain refining using COVERAL MTS 1582 through an MTS 1500 unit.



AFECO HEATING SYSTEMS, Kolhapur, Maharashtra, India.







The event was held at Vigyan Bhawan, New Delhi and was graced by the presence of Hon'ble Madam President Smt. Droupadi Murmu as the Chief Guest.

On behalf of AFECO HEATING SYSTEMS, the award was received by their Managing Director – Mr. Prakash Maladkar and Marketing Manager – Ms. Shivranjani Maladkar. The award was distributed by Hon'ble Shri R. K. Singh, Cabinet Minister – Ministry of Power, Government of India.

8



Porosity in ALuminium casting

C. Surianarayanan - Consultant, Email : c.surianarayanan@gmail.com

Internal Quality issues in aluminium casting and the reasons for it to be caused

Impregnation of aluminium castings for leak stopping is still considered the best option

Want to develop a method of improving faith in Aluminium high pressure die casting pressure tightness then it is better to concentrate on the details mentioned below.

Pressure die casting:

It has two basics in built by the process:

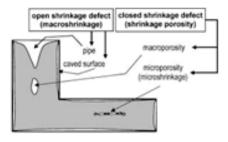
Toughness by work hardening due to the highpressure injection and filling

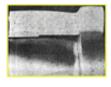
Packed tight with high velocity o that structure is clean with no deviations

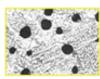
Porosity is caused by the absorption of nitrogen, oxygen and hydrogen in the molten weld pool which is then released on solidification to become trapped in the weld metal. Nitrogen and oxygen absorption in the weld pool usually originates from poor gas shielding

But This can be damaged due to the reason:

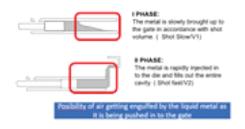
Product design with uneven section thickness which is not accommodable to flow the liquid alloy to get the part filled with packing against the shrinkage happening during the solidification. Shrinkage is an internal or external change in volume that occurs during a phase change in a metal's transition from a liquid state to a solid state at the exposed surface. This phenomenon occurs in processes like casting and concrete solidification

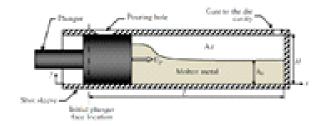






Air getting engulfed during the liquid alloy is getting compressed in the cold chamber before being pushed into the profile areas







Runner passages which are having stepped formation which can add void during the turbulence happening during the flow of the liquid alloy through the runner passages



Flow is disturbed by the sudden steps as marked and will create turbulence there by causing void in the liquid metal as it flows in to the profile



Flow is not disturbed and a smooth flow of the liquid metal is taken smoothly to the gate locations

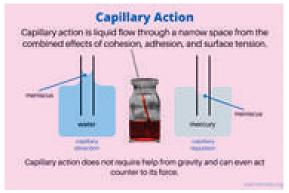
Improper selection of the fill ratio which can cause filling defects there by voids and shrinkage can happen

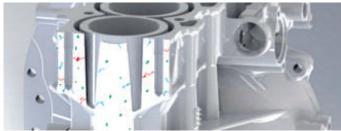
Air entrapment during the flow of the liquid alloy into the die from the runner to the final filling area of the profile can cause the internal gap in the profile sections

All these are the very basic but not taken into consideration during the Stages of Product design, Die engineering and the Process methodology

When there is an issue of void /shrinkage/ air entrapped in the casting it can lead to leak if the part is subjected to such applications. Where there is heat in the part due to the operational need's aluminium expands to the best possible.

This will create gap in the molecular and the expansion will lead the void to get the substance to leak out due to the capillary action





How do you reduce the porosity of aluminium casting?

Gas porosity can be eliminated through good mold design or by introducing nitrogen into the aluminum metal before the liquid pour.

Apart from these pints what are the other influential aspects that can cause the internal quality or the soundness of the casting results:

Thermal regulation of the die and the liquid metal has the highest degree of influence or effect on the filling pattern, packing pattern of the liquid metal.

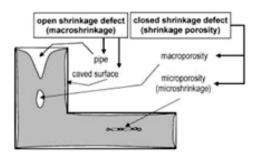
What is the best of the temperature for the alloy and the die?

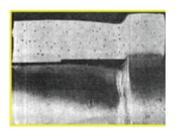
Alloy: Clean alloy with 650°C to 680°C maintained in the holding furnace will be the best. Because alloy temperature gets reduced in the transferring movement by the ladle from the furnace to the pouring spout of the shot sleeve. This is in regards with the conventional arrangement.

In the place of the dossing furnaces, it can be well within 650°C. Here the liquid metal transfer is happening in a closed passage and not transferred in the open air there by temperature loss is less anticipated.

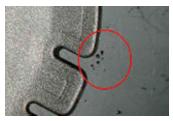
How can the temperature of the alloy influence or cause the probable internal defects?

Suppose the temperature is higher, then the liquides flow is faster than the solidus condition. This will flow the liquid metal faster to the profiles but before it is getting packed it may freeze on the areas where solid sections are to be filled with packing to avoid internal shrinkage porosity as shown in the sketch





If the temperature of the liquid metal is lower than the prescribed suggestions, then the flow will not be /may not be effective and likely chances of layer formation is possible. In this situation also liquid metal at a lower temperature may not flow to pack the part profiles. This will cause the internal quality issues like shrinkage, laminar flow etc.









Thermal managemnet of the cold chamber has its own ifluence on the liquid metal temperature. If the chamber is not maintained with the temperature of approximately 180°C to 250°C. It will draw the temperature of the molten alloy poured in to the chamberand reduce the alloy temperature. Hence its is suggested to have the water channels in the cold chamber and temperature be managed to the needs. Lower temperature of the liquid alloy will have iues with low silicon& higher ferrus content alloys.

How die coat water spray has the ifluence for this cause?

Earlier days oil or oil based coating was used just to protect the stesl being thermaly abused by the high temperature liquid metal being puhed with high velocity. Reaearchers say it goes anything up to 700°C as the metal squeezes through the gate. Even if the die is hot enough to with stand the enhanced heat

generated by each shot, it has to be controlled as well the steel has to be protected. Hence this water based soluble was invented and being ued till date.

Spray technique is related to the spray pattern , volume of water sprayed and ubsequent dry air being sprayed. There are likely chances of water getting stagnated in the profile pockets hence dry air spraying is recomanded for pushing the water out of the die and have the protection coating is there on the steel.

Supposing water is sprayed high then there are chances for the stagnation as well the die temperature being disturbed. This waill cause solidification defects, flow defects and stagnated water can explode by the hot alloy and gas can be formed which will get in to the casting profiles.

Buhler training teach the spray techniques along with the time line of the water as well the dry air. It is very clear that this also can cause the internal defects in the casting leave apart the other details such as runner, gate, temperature etc.

These are just a glimpse on the experinces has in all these years. It may not be scietific but true practical experience shared for the betterment of the future diecasters.



R. T. Kulkarni along with Mr. R. V. Apshankar visited Neo Wheels, Nashik



Brain Storming Meeting with Advisory Committee



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- Heat Treatment Furnaces
- Rotary Degassing Unit
- Density Index Unit

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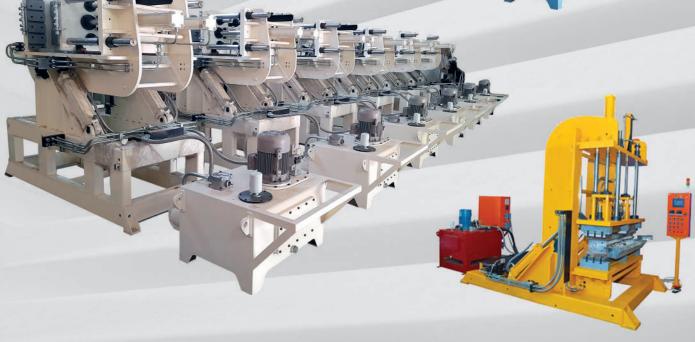
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- > Implementation of Industry 4.0



- Minda Industries Ltd has set up fully integrated manufacturing facility for Alloy wheel 2 wheeler.
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KEY FEATURES

- No need to remove Crucible
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- Increase crucible life
- Easy to remove & replace
- Heater replacement cost will be 40% due to new single leg Design
- Reduce Down time.
- Simple Design



Heater Replacement time							
Activity	Brick Lined Aluminium Melting furnace	Ceramic Insulated Aluminium Melting furnace 2 Leg Heater	New Design Aluminium Melting furnace 1 Leg Heater				
Cooling time	15 - 18 Hrs.	10-12 Hrs	No need to Cool down*				
Top Plate Removing time	15 Min	15-30 Mins	15-30 Mins				
Crucible Removal time	15-20 Min	15-20 Min	No need to remove crucible				
Failed Element identification	15-20 Min	15-20 Min	15-20 Min				
Element Replacement time for 1 Element	30-45 Min	30-45 Min	10-15 Min				
Crucible Installation Time	15-20 Min	15-20 Min	No need to remove crucible				
Top Plate Fixing time	15 Min	15-30 Mins	15-30 Mins				
Heating Time for 1st melt	4-6 Hrs	3-4 Hrs	0.5 - 1 Hrs.				
Total Down Time	17-20 Hrs.	15-19 Hrs.	1.5 - 2.6 Hrs.				
	Time saving compare	e to Brick lined furnace	15.5 - 17.4 Hrs.				
	Time saving compare	13.5 - 16.4 Hrs.					











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Report on Productivity Improvement

Theme: To Reduce Cycle Time at **VMC Machine in Machine Shop**

Team









Prashant Naik Babasaheb Satpute

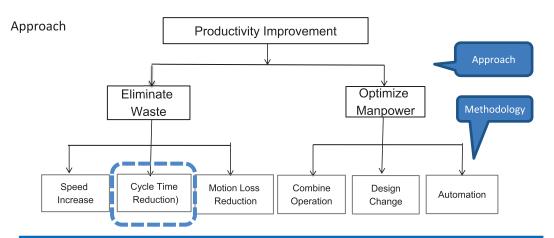


Sanjiv Nikam





Cycle Time Reduction Projects

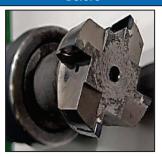


Total 19 Cycle Time Reduction Kaizen Identified in Machine Shop

Cycle Time Reduction Projects

Sr No	Item Number	Alt Item Number	Part Name	Machine type	Completion Date	Before Cycle time in Second	After Cycle time In Second	Improvement-%
1	WHSGSA01AG. B15 RSMT	00753-2330- 1102_REV:AG	Wireless Hsg	VMC	20-Mar-21	740	620	16
2	COVBK216 . B14 SMNS	A5E35758321_	216 Back cover	VMC	30-Mar-21	957	843	12
3	COVBXP208. B14 SMNS	A5E33602854	208 Back cover	VMC	30-Mar-21	848	770	9
4	MTRCOV01BK. B32 RSMT	03031-0096- 0001_ REV:BK	Cap-1	CNC	08-Apr-21	164	125	24
5	JNBOX01AH . B15 RSMT	00644-4193- 0001 REV:AH	Junction Box	VMC	11-Apr-21	343	278	19
6	HSGND9100. B14 MTSO	ND9100	METSO	VMC	15-Apr-21	755	590	22
7	HSGBS01 . B12 SMNS	A5E31454222A	SI Base Plate	VMC	26-Apr-21	946	720	24
8	AEBCOVTQ . B45 SMNS	A5E36028001_	Amber Blind cover	CNC	12-May-21	150	117	22
9	AEGCOVTQ . B45 SMNS	A5E36028014_	Amber Glass cover	CNC	28-May-21	178	135	24

Before



Four flute cutter

After



Six flute cutter

Before



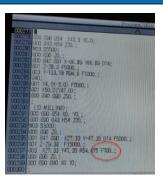
Air cutting feed 5000 mm/min

After



Air cutting feed 7500 mm/min

Before



- Cutting feed 700 mm/min
- Cutting feed 1000 mm/min

After



- Cutting feed 1015 mm/min
- Cutting feed 1500 mm/min

Results



Cost Saved - \$6K/Annum

Saved 45 Sec/Part 17% cycle time reduced

Kaizen 2 - Arm 1 Machining Cycle Time Reduction

Before



Boring & Chamfer tool required for Ø 63.9

After



PCD combination tool devolved for Ø 63.9 Boring finishing improved

Before



Boring & Chamfer tool required for Ø 35.014

After



PCD combination tool devolved for Ø 35.104 Boring finishing improved

Before



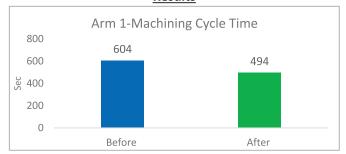
Boring & Chamfer tool required for Ø 26.1

After



PCD combination tool devolved for Ø 26.1 Boring finishing improved

<u>Results</u>



Cost Saved - \$17K/Annum

Saved 110 Sec/Part 18% cycle time reduced

Kaizen 3 - Washer Hinge Machining Cycle Time Reduction

Cutting feed 900 mm/min

CORREST CORRES

- Cutting feed 1000 mm/min
- Program modified



Air cutting feed 800 mm/min



- Air cutting feed 900 mm/min
- Program modified

Before



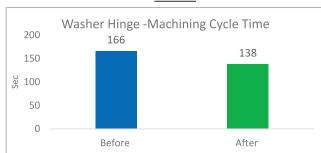
Axis movement time 18 Sec

After



- Eliminated axis movement
- Saved tool indexing time by 18 Sec

Results



Cost Saved - \$2.76K/Annum

Saved 28 Sec/Part 19% cycle time reduced

Kaizen 4 - Cycle time reduction - VMC / CNC Machine

Part Name :- Pokethrough

Machine - VMC

Operations - Drilling (Hinge & Handle)

Cycle time - 47 Sec



Major Improvements

- · Fixture modified
- Tools modified (Special combination tool)
- · OLD SPM modified with Servo Motor
- Program modified

Before

After

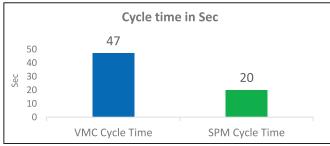


Machining operation on VMC



Machining operation converted on SPM

Results



Cost Saved - \$6K/Annum

Saved 27 Sec/Part 57% cycle time reduced

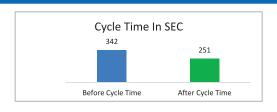
Kaizen 5 - Cycle time reduction - VMC / CNC Machine



Three tools are using for DIA 6.5, DIA 7.7 and Chamfer



Two different tools used for DIA 9.1 and DIA 5.4



After



Combine tool used for DIA 6.5, DIA 7.7 and Chamfer



Combine tool used for DIA 9.1 and DIA 5.4

Cost Saved - \$4.5K/Annum

Saved 91 Sec/Part 26% cycle time reduced

Kaizen 6 - Cutting Tools Modifications



- Carbide triangular insert used Tool Parts(180 life-2000 Meter)
- Cycle time-32Sec.

After



- PCD straight insert used
- Tool life-16000 Parts(1500 Meter)
- Cycle time-19Sec.

Before



- Carbide inserted thread mill used
- Tool life-1000 Parts(50 Meter)
- Cycle time -71Sec

After



- PCD thread mill used
- Tool life-1,00000 Parts(1500 Meter)
- Cycle time -47Sec



- 4 tool used for drilling, reaming, back chamfer & chamfer.
- Cycle time-80Sec



- Combination tool used for drilling, reaming, Back chamfer & chamfer.
- Cycle time-23Sec





- VCMT160404 PCD Insert with Holder Used
- Tool life-2047 Parts(40 Meter)
- Cycle time-32Sec/Part



- DCMW110304 PCD Insert with Holder
- Tool life-11569 Parts(226 Meter)
- Cycle time-20Sec/Part

Results

Total Saved \$179K/Annum



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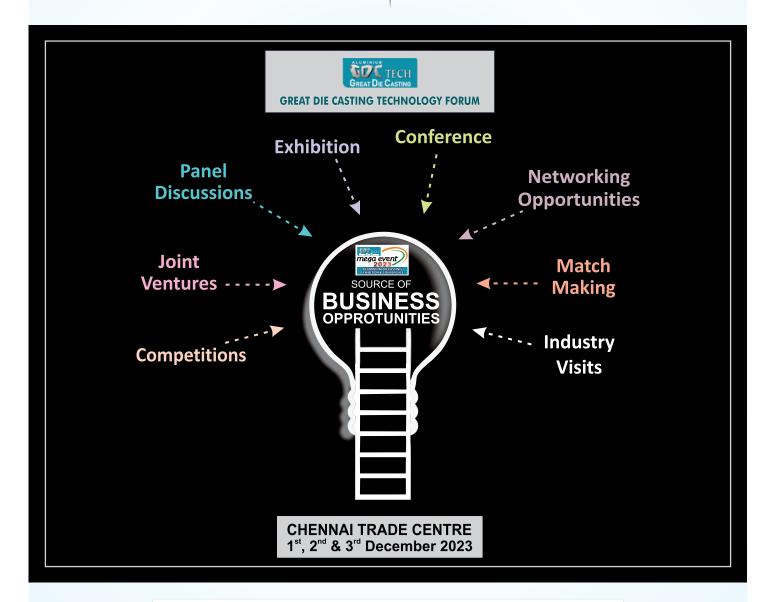








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